# **Characterization of the Structural and Surface Properties of a Synthesized Hydrous Hollandite by Gaseous Molecular Adsorption**

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The structural and surface properties of a hydrous hollandite-type manganese oxide  $(H-$ Hol) were examined by adsorption using different gases with molecular diameters below 0.4 nm.  $N_2$ ,  $Q_2$ , Ar, CO, and CO<sub>2</sub> with diameters above 0.33 nm are excluded from the tunnel structure of H-Hol, whereas  $H_2O$  and  $NH_3$  with diameters below 0.265 nm can be inserted into the structure. Disagreements were observed for the first and second run adsorption isotherms of  $H_2O$  and  $N\breve{H}_3$ , indicating that there is a strong interaction contributing to  $H_2O$ and  $NH<sub>3</sub>$  adsorption. Insertion of  $NH<sub>3</sub>$  into the tunnels of  $H-Hol$  is a very slow process; the rate-determining step is that of insertion into the inner  $H^+$  sites.  $H^+$  sites on H-Hol play an important role in both  $NH_3$  and  $H_2O$  adsorption. Adsorption and FT-IR results demonstrate that stepwise dehydration decreases the  $H^+$  sites available for  $NH_3$  adsorption, some but not all of which are recoverable by rehydration.

## **Introduction**

Microporous manganese oxides are one group of crystalline transition-metal oxides containing intratunnel or layered structures whose crystalline pore dimensions are estimated to be below 1 nm. Because of their potential application in rechargeable batteries,  $1-6$  ionsieving sorption, $7-12$  chemical sensors,  $13,14$  and cataly $sis, 6,15-19$  they have been actively studied in recent years. As one of the representative porous manganese

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oxides, hollandite-type manganese oxides (Hol) have a  $(2 \times 2)$  tunnel structure bordered by double chains of edge-shared  $MnO_6$  octahedra.<sup>6,20</sup> They can be either naturally occurring or artificially synthesized.6 Recently, Suib et al. have extensively studied the synthesis and structural and surface properties of hollandites and  $Cu^{2+}$ -, Fe<sup>2+</sup>-, and Ni<sup>3+</sup>-substituting hollandites with various techniques such as X-ray diffraction (XRD), transmission electron microscopy (TEM), secondary electron microscopy (SEM), temperature-programmed desorption (TPD), and so on.3,6,21-<sup>25</sup> The electrochemical and ion-sieving properties of hollandites have also been widely studied for several decades.<sup>7-9,12</sup> However, despite great expectations as a new type of molecular sieve, research reports on its gas adsorptivity are limited. Knowledge regarding its effective pore dimension available for gas probing has not been completely proven either.

Figure 1 shows a schematic projection of hollandite structure in the (*ab*) plane. One can take a rectangular unit cell with a common size for *a* and *b*, whose edges intersect every external corner of  $MnO_6$  octahedra surrounding the  $(2 \times 2)$  tunnel.<sup>21,22</sup> By subtracting the thickness of the Mn layer from the interlayer space, the crystalline pore opening can be estimated as 0.46 nm. However, it is common knowledge that a crystalline pore opening is not always the same as the effective pore opening because the latter is influenced not only by

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Unit cell:  $a = b = 0.981$  nm,  $c = 0.286$  nm

**Figure 1.** Projection of the  $(2 \times 2)$  tunnel structure of hollandite-type manganese oxide in the (*ab*) plane.

accommodation of framework ions but also by the presence of intratunnel cations. For example, accommodation of intratunnel cations with ionic size from 0.07  $(Li^+)$  to 0.16 nm  $(Cs^+)^{12}$  in the one-dimensional tunnel of hollandite may restrict the accessibility of their tunnels to gaseous molecules. According to O'Young et al.,26 the microporous structure of hollandite-type manganese oxides cannot be satisfactorily elucidated by lowtemperature argon adsorption. The best way to probe the pore size distribution of a microporous system is the molecular probing method using gas molecules with different dimensions.<sup>27,28</sup> In a previous study, we reported that  $NH<sub>3</sub>$  can be inserted into the tunnel structure of a hydrous hollandite-type manganese oxide.<sup>29</sup> In the present study, we examined the micropore system of hollandite by adsorption using common molecules with molecular diameters below 0.4 nm. Emphasis has been given to adsorptivity of small  $NH_3$  and  $H_2O$ molecules.

# **Experimental Section**

**Material and Characterization.** To probe the microporous structure of Hol without the influence of cations, hydrous hollandite (H-Hol) was used in this research because the proton ionic size can be ignored. H-Hol was synthesized by the hydrothermal route reported by Thackeray et al.<sup>2</sup> Commercial powders of *γ*-MnOOH (Tosoh Co., Manganite) and Li<sub>2</sub>- $CO<sub>3</sub>$  (Wako Pure Chemical Industrial Ltd., high purity) were mixed in a crucible and calcined at 1073 K for 48 h to form a Li<sub>2</sub>MnO<sub>3</sub> precursor. The Li<sub>2</sub>MnO<sub>3</sub> precursor was then added to a 3 M sulfuric acid solution in a Teflon-sealed autoclave, allowing hydrothermal reaction at 363 K for 48 h. The H-Hol powders produced were dried at 343 K for one night and kept in a desiccator before further use. The chemical analysis result showed that the prepared product has a unit cell chemical formula of  $HMn_8O_{16}$ ; the average atomic valence for Mn is 3.88. The X-ray diffraction (XRD) pattern of the prepared H-Hol was determined by a Rigaku RINT 1200 diffractometer from 10 to 70° at a scanning speed of 0.02°/min in a chamber with humidity and temperature control. The thermal gravimetry (TG) and differential thermal analysis (DTA) of the prepared <sup>H</sup>-Hol were carried out with a TG-DTA 2000 system (Mac Science Co. Ltd.) at a ramp rate of 10 K/min. Commercial  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> was used as a standard material for TG/DTA measurements.

**Adsorption Measurement.** The adsorption of  $N_2$ ,  $Q_2$ , Ar, CO,  $CO<sub>2</sub>$ , H<sub>2</sub>O, and NH<sub>3</sub> was carried out by a volumetric method using commercial volumetric apparatus (Autosorp-1 of Quanta-Chrom Co. or Belsorp 18 of Japan Bell Co.). The adsorption temperatures are 77 K for  $N_2$ ,  $O_2$ , Ar, and CO, 195 K for  $CO<sub>2</sub>$ , and a temperature in the range of 278-318 K for H2O and NH3. Prior to adsorption, the samples were evacuated under 1 mPa at 393 K (or 573 K) for 2 h. After completion of a first-run adsorption, H2O or NH3-adsorbed H-Hol was evacuated under 10 mPa at the adsorption temperature for at least 2 h and then subjected to a second-run adsorption at the same temperature. The amount of irreversible  $NH<sub>3</sub>$  and H2O adsorption was obtained by subtracting the second-run adsorption isotherm from the first. The relationship of  $H_2O$ and NH<sub>3</sub> adsorption on H-Hol to pre-evacuation temperature (393, 473, or 573 K) was also examined. The NH<sub>3</sub> or H<sub>2</sub>O adsorption on  $H_2O$ - or  $NH_3$ -preadsorbed H-Hol was carried out after evacuating the sample preadsorbed by  $H_2O$  for 2 h or NH3 for 30 min at the same adsorption temperature. The equilibrium time for obtaining every point of adsorption isotherm is 300 s for  $NH<sub>3</sub>$  and 500 s for the other gases. Commercial gases with purities greater than 99.99% were used for adsorption measurements without further purification.  $H_2O$ was purified by the freeze-pump-thaw method before use.

**FT-IR Spectroscopic Measurement.** The NH<sub>3</sub> surface adspecies and the change of surface sites on H-Hol for NH3 adsorption were observed by FT-IR spectroscopy. A special quartz cell mounted with a pair of NaCl windows to allow penetration of an IR beam through the centers of the windows was connected to a vacuum line for sample pre-evacuation and introduction of gas without exposure to the atmosphere. Prior to measurements, H-Hol was pressed at 250 kPa to form a thin wafer of 1 cm diameter, which was placed in a quartz holder suspended by means of a Pt wire in the quartz cell. After evacuation at 393 K for 15 min (condition a) at the upper part of quartz cell, the H-Hol wafer was shifted down to the measurement position in a room temperature environment  $(295-297)$  K). NH<sub>3</sub> was then introduced and maintained at equilibrium pressures of 1.0 (condition b) and 3.6 kPa (condition c) for 15 min each. After the NH3-adsorbed H-Hol was successively evacuated at 393 (condition d) and 473 K (condition e) for 15 min each, the sample was subjected to another NH3 adsorption at an equilibrium pressure of 3.2 kPa for 15 min (condition f) and then evacuated at 473 (condition g) and 573 K (condition h), for 15 min each. After that,  $NH<sub>3</sub>$  adsorption at room temperature and desorption at 573 K were repeated four times with an NH3 equilibrium pressure of 1.6, 1.1, 1.2, or 1.0 kPa and at an adsorption time of 15 min each (Conditions i, j, k, l, m, n, o, and p sequentially represent each of the adsorption and desorption steps). Finally, the sample was evacuated at 573 K for 25 min (condition q), and at 773 (condition r) and 973 K (condition s) for 10 min each. The IR absorbance corresponding to each of the above steps was collected from 32 scans at a resolution of 2 cm<sup>-1</sup>. NH<sub>3</sub> gas of 99.999% purity was used for IR measurements after further purification by a freeze-pump-thaw method.

## **Results and Discussion**

**Sample Characterization and Adsorption of Various Gaseous Molecules on H**-**Hol.** XRD measurements confirm that the prepared H-Hol has a structure similar to natural cryptomelane, 30,31 having a tetragonal unit cell of  $a = b = 0.981$  nm and  $c = 0.286$ nm (Figure 1). Figure 2 shows the change in XRD patterns of the prepared H-Hol after evacuation at

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**Figure 2.** XRD patterns of (a) as-prepared H-Hol and those after evacuation for 2 h at (b) 373, (c) 473, (d) 573, (e) 673, and (f) 773 K.



**Figure 3.** TG/DTA curves of the as-prepared H-Hol. Condition: ramp rate, 10 K/min; curtain gas, air (100 mL/min).

various temperatures for 2 h. The prepared H-Hol basically resists temperature-induced changes up to and including 673 K although the diffraction peaks were broader and of less intensity with evacuation at 673 K. Dehydration at 773 K leads to the transformation of hollandite to an  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> crystal. The TG/DTA curves of the as-prepared H-Hol in Figure 3 show the result corresponding to the XRD pattern changes in Figure 2, giving a gradual dehydration until 673 K before the first endothermic peak due to the crystal transformation to  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub>. The second crystal transformation from  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> to Mn<sub>3</sub>O<sub>4</sub> gives rise to a second step in the TG curve and a great endothermic peak in the DTA curve at around 1200 K. The extent of dehydration, 6.2 wt %, between 303 and 673 K is much greater than that released by detachment of the  $H^+$  contents in  $H$ -Hol (∼1.3 wt %), indicating that one part of water can be



**Figure 4.** Adsorption isotherms of  $N_2$  (0),  $O_2$  (0), Ar ( $\blacktriangle$ ), and CO ( $\triangle$ ) at 77 K, CO<sub>2</sub> ( $\blacktriangledown$ ) at 195 K, NH<sub>3</sub> first run ( $\blacksquare$ ), NH<sub>3</sub> second run  $(\square)$ , and the difference between first and second runs ( $\Box$ ) at 298 K, and H<sub>2</sub>O first ( $\blacklozenge$ ) and second ( $\Diamond$ ) runs at 298 K, and  $CO<sub>2</sub>$  desorption isotherm at 195 K ( $\triangledown$ ).

retained on the surface of H-Hol until high temperature.

Figure 4 shows the adsorption isotherms of Ar and CO in comparison with those of  $N_2$  and  $O_2$  at 77 K,  $CO_2$ adsorption isotherm at 195 K, the first- and second-run adsorption isotherms of  $H_2O$  and  $NH_3$  at 298 K, and an irreversible adsorption isotherm of  $NH<sub>3</sub>$  at 298 K on <sup>H</sup>-Hol. All the adsorption isotherms except that of NH3 show typical type II shapes, which are characteristic of multilayer condensation on a flat surface.<sup>27</sup> No hysteresis was observed in the desorption branches of Ar, CO,  $N_2$ , and  $O_2$  (not shown for clarity), indicating that no mesopores were present as probed by these molecules in H-Hol. However, a slight desorption hysteresis became evident for  $CO<sub>2</sub>$  having a smaller molecular diameter  $(0.33 \text{ nm}^{37})$ . This desorption hysteresis persists to the lower  $P/P_0$  value ( $P/P_0 < 0.1$ ), demonstrating the existence of activated adsorption and stronger inclusion of  $CO<sub>2</sub>$  in the micropores of H-Hol. The first-run adsorption isotherm of  $NH<sub>3</sub>$  showed a sharp uprising at  $P/P_0$  < 0.01 and a gradual increase after  $P/P_0$  = 0.01. The sequence of adsorption amount at  $P/P_0 = 0.3$  on H-Hol is  $NH_3$  (first-run) > H<sub>2</sub>O (first-run) > O<sub>2</sub> > N<sub>2</sub>  $= Ar > CO = CO<sub>2</sub>$ . NH<sub>3</sub> and H<sub>2</sub>O adsorption amounts are 2-3-fold greater than those of other gases. The disagreement between the first- and the second-run adsorption of  $NH<sub>3</sub>$  and  $H<sub>2</sub>O$  implies the existence of a strong interaction of  $NH<sub>3</sub>$  and  $H<sub>2</sub>O$  with the surface of  $H$ –Hol. Formation of N $H_4^+$  by N $H_3$  adsorption<sup>29</sup> leads<br>to a greatly enhanced irreversible adsorption at low to a greatly enhanced irreversible adsorption at low relative pressure. A comparatively slight irreversible H2O adsorption is ascribed to either the stronger H-bond interaction or the possible formation of  $H_3O^+$  species, which have been confirmed from the lattice expansion phenomenon of H-Hol after  $H<sub>2</sub>O$  adsorption by XRD measurements. To make parallel comparisons, the Brunauer-Emmett-Teller (BET) equation was applied for all isotherms to calculate the specific surface areas,  $S_{\text{BET}}$ , despite possible involvement of chemisorption in case of  $NH_3$  and  $H_2O$ . As shown in Table 1, the  $S_{BET}$ values from Ar, CO,  $CO_2$ , N<sub>2</sub>, and  $O_2$  adsorption are very similar to one another but smaller or much smaller than

**Table 1. Surface Area of H**-**Hol Calculated from Adsorption of Each Gas**

adsorbate	evacuation T/K	d∕nm	adsorption T/K	$S_{\rm BET}/m^2/g$	$S_{\rm t. total}$ <sup>a</sup> /m <sup>2</sup> /g	$S_{\text{text}}^{\text{a/m2/g}}$	$S_{\text{t.int}}^{\text{a/m2/g}}$
$N_2$	393	0.364		73	68	56	12
	573			79			
O <sub>2</sub>	573	0.346	77	71	nd	nd	nd
Ar	573	0.340		79	nd	nd	nd
CO	573	0.376		67	nd	nd	nd
CO <sub>2</sub>	573	0.330	195	64	nd	nd	nd
$H2O$ (first run)	393	0.265	298	95	nd	nd	nd
$NH3$ (first run)	393	0.26	298	161	nd	nd	nd

*<sup>a</sup>* nd: Not determined.



Figure 5. (a) First- and (b) second-run NH<sub>3</sub> adsorption isotherms and (c) first- and (d) second-run  $H<sub>2</sub>O$  adsorption isotherms on H-Hol at 298 K after preevacuation at 393  $\left(\bullet\right)$ , 473 ( $\blacksquare$ ), and 573 K ( $\blacktriangle$ ). Unfilled symbols represent the desorption branch of the adsorption with the same type of symbol.

those from  $H_2O$  (first-run) or  $NH_3$  (first-run) adsorption. By analyzing the adsorption isotherms using a de Boer *t*-plot,<sup>32</sup> it is possible to know the internal and external surface areas. On the basis of a standard  $N_2$  adsorption isotherm on nonporous alumina, $27,32$  we obtained from the  $N_2$  adsorption isotherm the total, the external, and the internal surface areas, denoted as  $S_{t,\text{total}}$ ,  $S_{t,\text{ext}}$ ,  $S_{t,\text{int}}$ , respectively. As shown in Table 1, the *S*t,ext values from  $N_2$  adsorption are very close to those of  $S_{t,total}$  and  $S_{BET}$ values. It thus follows that not only  $N_2$  but also  $O_2$ , Ar, CO, and  $CO<sub>2</sub>$  are excluded from the microporous tunnels of H-Hol. The greater  $S_{\text{BET}}$  values from NH<sub>3</sub> and H<sub>2</sub>O adsorption demonstrate that the two molecules can be inserted into the tunnel structure although it is hard for the inserted  $NH_3$  and  $H_2O$  to be desorbed because of strong localization. Although H-Hol has a crystalline pore opening of 0.46 nm, its effective pore opening must be less than 0.33 nm (the dynamic diameter,  $d$ , of  $\text{CO}_2{}^{37}\text{)}$ and close to 0.265 nm (the *d* value of  $H_2O^{37}$ ).

**Relationship of NH3 and H2O Adsorption with the Extent of Dehydration of H**-**Hol.** Figure 5 shows the first- and second-run adsorption isotherms of NH3

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**Table 2. NH<sub>3</sub> Adsorption Amount at**  $P/P_0 = 0.027$ **,**  $V_{NH_3,5}$ **, and Irreversible Surface Area,** *S***irr, under Various Conditions**

evacuation T/K	adsorption T/K	first- run	second- run	irrev. (Langmuir)	$S_{\rm irr}$ $m^2/g$
393	278	40	13	27(27)	91
	298	41	11	30(31)	104
	318	42	13	29 (34)	114
473	278	29	12	17 (18)	60
	298	34	9	25(25)	84
	298a	38	13	25(25)	b
	318	35	11	24 (24)	81
573	278	16	9	7 (7)	24
	298	15	8	7 (6)	20
	298 <sup>a</sup>	26	10	16 (16)	b
	318	16	8	8(7)	25

*<sup>a</sup>* After H2O preadsorption. *<sup>b</sup>* Not determined.

and  $H_2O$  at 298 K on H-Hol after dehydration at various temperatures. Both the first- and second-run adsorption amounts of NH<sub>3</sub> decreased with increase of dehydration temperature, demonstrating that  $H^+$  content in H-Hol plays an important role in  $NH<sub>3</sub>$  adsorption. The smaller difference among the second-run adsorption amounts of  $NH<sub>3</sub>$  at various dehydration temperatures indicates that protons contained in H-Hol have a great responsibility for irreversible  $NH<sub>3</sub>$  adsorption. Table 2 shows the NH<sub>3</sub> adsorption amounts at 298 K on H-Hol (at  $P/P_0 = 0.027$ ),  $V_{NH3, s}$ , at various dehydration temperatures. The  $NH<sub>3</sub>$  irreversible adsorption amount on H-Hol at a dehydration temperature of 573 K is less than one-fourth of that at a dehydration temperature of 393 K. Since it is not easy to apply an adsorption theory to analysis of  $NH<sub>3</sub>$ adsorption isotherms, which contain contributions from both chemisorption and physical adsorption, we only fit the irreversible  $NH<sub>3</sub>$  adsorption isotherms with the Langmuir equation (not shown for clarity), whereas the surface area for irreversible NH3 adsorption, *S*irr, was calculated from the Langmuir saturated adsorption amount by tentatively using the  $NH<sub>3</sub>$  molecular sectional area  $(0.129 \text{ nm}^2)$ .<sup>33</sup> As shown in Table 2, the Langmuir saturated adsorption amount of NH3 matches well with  $V_{NH_3s}$ . The  $S_{irr}$  values at dehydration temperatures of 393 and 473 K are much greater than the  $S_{\rm BET}$ values from Ar,  $N_2$ ,  $O_2$ , CO, and  $CO_2$  adsorption, revealing a great contribution of chemisorption during  $NH<sub>3</sub>$  adsorption.

On the other hand, while the first-run adsorption isotherms of  $H<sub>2</sub>O$  for dehydration at 393 and 473 K overlap with each other, the second-run adsorption for

<sup>(33)</sup> Young, D. M.; Crowell, A. D. *The Physical Adsorption of Gases*; Butterworth : London, 1962.

**Table 3. H2O Adsorption Area under Various Conditions**

evacuation	adsorption	$S_{BET,H_2O}/m^2/g$			
T/K	77K	first-run	second-run	irrev.	
393	283	71	53	17	
	298	91	61	30	
	298 <sup>a</sup>	50	h	h	
	308	100	73	26	
473	298	92	55	37	
	298a	55	Ь	h	
573	283	74	46	28	
	298	68	44	23	
	308	75	48	26	

*<sup>a</sup>* After NH3 preadsorption. *<sup>b</sup>* Not determined.

dehydration at 473 K is slightly smaller than that for dehydration at 393 K, indicating that the dehydrated sites at 473 K are recoverable on readsorption of  $H_2O$ possibly due to dissociative adsorption. In contrast to the case of  $NH<sub>3</sub>$  adsorption, the reduction in the firstrun  $H_2O$  adsorption amount between dehydration temperatures of 393 and 573 K is almost the same as that in the second-run adsorption amount, indicating that high-temperature dehydration has a smaller effect on the reduction of irreversible adsorption of  $H<sub>2</sub>O$ . Table 3 shows the  $S_{BET, H_2O}$  values from  $H_2O$  adsorption at 298 K after dehydration at various temperatures. The  $S_{BET,H_2O}$  values from the second-run  $H_2O$  adsorption isotherms at dehydration temperatures of 573 K are even smaller than the  $S_{t,ext}$  values from  $N_2$  adsorption, reflecting the important role of surface  $H^+$  sites on  $H$ –Hol in  $H_2O$  adsorption. Since the crystal structure is not greatly changed by dehydration until 673 K, as shown by the XRD result, the tunnel porous structure of H-Hol may not significantly change until 673 K. The small  $H<sub>2</sub>O$  adsorption at the dehydration temperature of 573 K implies the disappearance of  $H<sub>2</sub>O$  dissociative adsorption sites due to a partial rearrangement of the <sup>H</sup>-Hol surface by certain valence transformation of surface Mn ions ( $Mn(IV) \rightarrow Mn(III)$ ).

Figure 6 shows the  $NH<sub>3</sub>$  and  $H<sub>2</sub>O$  adsorption isotherms at 298 K on  $H<sub>2</sub>O-$  and  $NH<sub>3</sub>-preadsorbed H–Hol,$ respectively. Preadsorption of  $H<sub>2</sub>O$  on H-Hol dehydrated at 473 K slightly enhanced the weak  $NH<sub>3</sub>$  adsorption, leading to the second-run NH<sub>3</sub> adsorption amount which is the same as that at dehydration temperatures of 393 K, but had almost no effect on irreversible  $NH<sub>3</sub>$ adsorption. This result correlates with the above results, in which dehydration at 473 K leads to detachment of one part of weak  $H^+$  sites on  $H$ -Hol, which can be reversibly recovered. Readsorption of  $H_2O$  produces weak  $H^+$ sites by dissociative adsorption for  $NH<sub>3</sub>$  adsorption, whereas one part of the strong  $H^+$  sites is irreversibly lost by surface rearrangements due to Mn ion valence changes at dehydration temperatures of 473 K. On the other hand, preadsorption of H<sub>2</sub>O on H-Hol dehydrated at 573 K did not increase the weak NH<sub>3</sub> adsorption (the second-run adsorption) while largely enhancing the irreversible  $NH<sub>3</sub>$  adsorption, indicating that, by dehydration at 573 K, the weak  $H^+$  sites disappear by surface rearrangements while one part of the dehydrated sites from strong  $H^+$  sites can still be recovered by readsorption of  $H_2O$ . In addition, the amount of irreversible  $NH<sub>3</sub>$ adsorption on  $H<sub>2</sub>O$ -preadsorbed H-Hol after dehydration at 573 K is still smaller than that on H-Hol dehydrated at 473 K, implying that one part of the strong NH<sub>3</sub> adsorption sites is also irreversibly lost by



Figure 6. Comparison of NH<sub>3</sub> adsorption isotherms at 298 K on fresh and  $H_2O$  preadsorbed H-Hol after evacuation at (a) 473 and (b) 573 K, respectively, and those of  $H_2O$  at 298 K on fresh and NH<sub>3</sub> preadsorbed H-Hol after evacuation at (c) 393 and (d) 473 K, respectively.  $\blacksquare$ ,  $\Box$ ,  $\Box$ : first-run, second-run, and irreversible adsorption without  $H_2O$  or  $NH_3$  preadsorption, respectively.  $\bullet$ ,  $\circ$ ,  $\circ$ : first-run, second-run, and irreversible adsorption with H<sub>2</sub>O or NH<sub>3</sub> preadsorption, respectively.

dehydration at 573 K. When  $NH<sub>3</sub>$  was preadsorbed on H-Hol dehydrated at 393 K, the  $H_2O$  adsorption amount was smaller than that of the second-run adsorption without NH<sub>3</sub> preadsorption. This decrease in  $H_2O$  adsorption is ascribed to  $NH_3$  occupying weak  $H^+$ sites, which will be lost by dehydration at 473 K and is reproducible by  $H<sub>2</sub>O$  readsorption, because the decrease in the second-run  $H_2O$  adsorption by  $NH_3$  preadsorption on H-Hol dehydrated at 393 K (11 m<sup>2</sup>/g from Table 3) is similar to twice the difference of  $S_{\text{BET,H}_2\text{O}}$  from the second-run H2O adsorption at dehydration temperatures between 393 and 473 K (6 m<sup>2</sup>/g from Table 3). However, the  $H_2O$  adsorption isotherm on H-Hol preadsorbed by NH3 overlaps very well with the secondrun  $H_2O$  adsorption without  $NH_3$  preadsorption on <sup>H</sup>-Hol dehydrated at 473 K. This phenomenon suggests that  $H_2O$  cannot be inserted into the innermost  $H^+$  sites in the tunnel where  $NH<sub>3</sub>$  can be strongly adsorbed.

**Relationship of NH3 and H2O Adsorption with** Adsorption Temperature. Figure 7 shows NH<sub>3</sub> and H2O adsorption isotherms on H-Hol at various adsorption temperatures. Increasing the adsorption temperature in the range of 278-318 K gradually increased not only the first-run but also the second-run  $NH<sub>3</sub>$  adsorption, indicating that even second-run  $NH<sub>3</sub>$  adsorption is partially specific. Calorimetric results show that the initial second-run  $NH<sub>3</sub>$  adsorption heat can be as high as  $80-90$  kJ/mol,<sup>34</sup> much more than the condensation heat of NH<sub>3</sub> (19.9 kJ/mol at 298 K<sup>35</sup>), reflecting its property of specific adsorption. On the other hand, while

<sup>(34)</sup> Wang, Z.-M.; Kanoh, H. unpublished results.

<sup>(35)</sup> *Handbook of Chemistry and Physics*, 76th ed.; David, R. L., Ed.; CRC Press: Boca Raton, 1995; pp 6-116.



**Figure 7.** Adsorption isotherms for NH<sub>3</sub> at 278 ( $\bullet$ ,  $\circ$ ), 298  $(\blacksquare, \square)$ , and 318 K  $(\blacktriangle, \triangle)$  and those for H<sub>2</sub>O at 283  $(\blacklozenge, \diamond)$ , 298  $(\blacksquare, \square)$ , and 308  $(\blacktriangledown, \triangledown)$  K on H-Hol after evacuation at (a) 473 and (b) 573 K for NH<sub>3</sub> and (c) 473 and (d) 573 K for H<sub>2</sub>O. The solid and unfilled symbols represent first- and second-run adsorption, respectively.

the same specific properties of the first- and second-run H2O adsorption were observed on H-Hol dehydrated at 473 K, the  $H_2O$  adsorption on H-Hol dehydrated at 573 K did not change significantly with adsorption temperature. This result coincides with those in the above section, further indicating that weak  $H^+$  sites disappear by dehydration at 573 K and that the innermost strong  $H^+$  sites are not accessible to  $H_2O$ adsorption.

As shown in Tables 2 and 3, the irreversible adsorption of  $NH<sub>3</sub>$  and  $H<sub>2</sub>O$  basically increased with increase of adsorption temperature, indicating that adsorption of  $NH_3$  or  $H_2O$  may be controlled by an activation adsorption to expand the lattice structure due to formation of larger  $NH_4^+$  or  $H_3O^+$  as well as diffusion resistance along the tunnel pore structure. Figure 8 shows the time-courses of first-run  $NH<sub>3</sub>$  adsorption at 283 and 298 K on H-Hol dehydrated at 393 K. A common adsorption rate was observed at the initial adsorption stages  $(t \leq 100 \text{ s})$  for the two adsorption temperatures. However, a difference in adsorption rate became apparent after  $t = 100$  s. The higher adsorption  $temperature$  accelerates  $NH<sub>3</sub>$  adsorption, resulting in a higher adsorption amount with time. The change in the time course of  $NH<sub>3</sub>$  indicates that two sub-stages exist for  $NH_3$  adsorption on H-Hol: one is the initial stage occurring at the external surface or near the entrance, and the other is the  $NH<sub>3</sub>$  insertion stage into the intratunnel structure, which is the more ratedetermining process.

**FT-IR Observation of NH3 Adsorption Sites on Repeated Adsorption**-**Desorption Cycling.** Figure 9 shows the  $NH<sub>3</sub>$  adspecies in the range around 1400  $cm<sup>-1</sup>$  under various conditions (the detailed description on the conditions is given in the Experimental Section).



Figure 8. Time-courses of first-run NH<sub>3</sub> adsorptions at 283 (O) and 298 ( $\square$ ) K with Pe = 2 kPa on H-Hol dehydrated at 393 K.

After evacuation of H-Hol at 393 K (condition a), a small sharp peak at  $1420 \text{ cm}^{-1}$  was observed. Because this band always appeared after the same procedure for several different batches of H-Hol samples, we suppose that it came from some contaminants, e.g., some adsorbed NH $_4^+$  species or other adspecies. NH $_3$  adsorption on H-Hol dehydrated at 393 K (condition b) gave a sharp peak of great intensity at  $1400 \text{ cm}^{-1}$ , which can be ascribed to an  $\mathrm{NH_4}^+$  species. $^{36}$  This peak was not seriously enhanced by rising NH<sub>3</sub> gas pressure (condition c) but gradually decreased by evacuation at 393 and 473 K (conditions d and e). The position of  $\mathrm{NH}_4^+$  species was constant after desorption at 393 K but shifted to a higher band position after that at 473 K. NH<sub>3</sub> readsorption (condition f) on H-Hol successively evacuated at 473 K again gave a sharp peak at  $1400 \text{ cm}^{-1}$  and subsequent desorption (condition g) at 473 K retained the same peak shapes and positions as that of condition e. Thus, NH<sub>3</sub> adsorption sites to form NH $_4^+$  species are not changed and  $NH<sub>3</sub>$  can be repeatedly adsorbed on <sup>H</sup>-Hol at a dehydration temperature below 473 K although the irreversible  $NH<sub>3</sub>$  adspecies cannot be completely desorbed. However, NH<sub>3</sub> adsorption on H-Hol successively evacuated at 573 K leads to only a slight increase in the peak at  $1420 \text{ cm}^{-1}$  instead of a peak at  $1400 \text{ cm}^{-1}$  (condition i). A peak at  $1600 \text{ cm}^{-1}$ , which can be assigned to an  $NH<sub>3</sub>$  adspecies on Lewis acid sites,  $36$ also concurrently appeared. With the increase in the number of repeated cycling of NH<sub>3</sub> adsorption-desorption (at 573 K), the peak at 1420  $cm^{-1}$  gradually decreased in intensity in concurrence with the gradual increase in intensity of the peak at  $1600 \text{ cm}^{-1}$  and finally disappeared (conditions  $j-p$ ). Thus, the change in NH3 adsorption sites on repeated adsorption-desorption (at 573 K) reflects the gradual diminishing of  $H^+$ sites on H-Hol and their transformation to Lewis acid sites, suggesting easy rearrangement of the H-Hol surface at a dehydration temperature of 573 K. The appearance of two  $-NH_4^+$  adspecies implies the pres-

<sup>(36)</sup> Tsyganenko, A. A.; Pozdnyakov, D. V.; Filimonov, V. N. *J. Mol. Struct.* **1975**, *29*, 299.



Figure 9. FT-IR observation on repeating NH<sub>3</sub> adsorption-vacuum evacuation. Conditions (a-s) are the same as those described in the Experimental Section.

ence of two  $NH<sub>3</sub>$  adsorption sites on H-Hol: the comparatively weak  $H^+$  sites exist in large amounts but disappear on evacuation above 573 K, and the stronger, less plentiful  $H^+$  sites can be retained at a higher temperature. As the result of sufficient dehydration, surface rearrangement, and partial or complete destruction of the hollandite structure, the  $NH<sub>3</sub>$  adsorption sites are almost entirely lost and  $NH<sub>3</sub>$  adspecies become obscure by evacuation above 673 K (conditions q to s).

**Schematic Model for the Structural and Surface Properties of H**-**Hol and Their Changes on Increasing Temperature.** From the above adsorption results, a schematic model of the pore structure of <sup>H</sup>-Hol for gas adsorption is shown in Figure 10a. Although the crystalline pore opening of H-Hol is estimated as 0.46 nm,21,22 we believe that this pore opening is the distance between the centers of oxygen ions of the two neighboring  $MnO_6$  octahedra on each side of the  $(2 \times 2)$  tunnel. The dimension of 0.216 nm obtained by subtracting the diameter of  $O^{2-}$  ions (0.244) nm for a three coordination number<sup>35</sup>) from 0.46 nm is small enough to exclude Ar,  $N_2$ ,  $O_2$ ,  $CO$ , and  $CO_2$  with dynamic molecular diameters between 0.33-0.364 nm<sup>37</sup> from the tunnel. Although the value of 0.216 nm depends on the  $O^{2-}$  radius value in the literature, it is rather close to the dynamic diameters of  $H_2O$  and  $NH_3$  $(H<sub>2</sub>O, 0.265$  nm; NH<sub>3</sub>, 0.26 nm<sup>37</sup>). This may explain why adsorption, especially physical adsorption of  $H_2O$  and NH3, into the tunnel is not easy but requires the aid of stronger irreversible adsorption. Hence, adsorption isotherms for  $NH<sub>3</sub>$  and  $H<sub>2</sub>O$  previously shown may not reflect a real equilibrium adsorption but a quasiequilibrium adsorption. The difference in  $H_2O$  and  $NH_3$ adsorption can be ascribed either to a thermodynamic disadvantage due to the weaker specific interaction or a higher activation energy to form  $H_3O^+$  than to form  $\mathrm{NH_4^+}$  or to a larger diffusion resistance along the tunnel



 $(b)$ 



**Figure 10.** (a) Molecular adsorption schematic of  $N_2$ ,  $O_2$ , Ar, CO,  $CO<sub>2</sub>$ , H<sub>2</sub>O, and NH<sub>3</sub> on H-Hol and (b) dehydration schematic of H-Hol with increasing temperature.

structure due to a larger molecular size of  $H<sub>2</sub>O$  in comparison with  $NH<sub>3</sub>$ . Strong acid-base interactions give NH3 molecules great accessibility to the innermost H<sup>+</sup> sites with strong adsorption energy. Formation of  $\mathrm{NH}_4^+$  due to  $\mathrm{NH}_3$  adsorption brings about expansion of the lattice structure,<sup>29</sup> leading to  $NH<sub>3</sub>$  being more easily transported along the one-dimensional tunnel.

Figure 10b shows a schematic model describing the dehydration process of H-Hol. As the dehydration amount of H-Hol from room temperature to 673 K is much larger than the theoretical H2O amount from the (37) Breck, D. W. *Zeolite Molecular Sieves* - *Structure, Chemistry,*

*and Use*; Wiley-Interscience: New York, 1974.

lattice protons, it seems from the  $H<sub>2</sub>O$  and  $NH<sub>3</sub>$  adsorption results that lattice protons desorb sequentially from weaker to stronger intensity protons together with included water. Dehydration of H-Hol is unlike that in other metal oxides such as silica, alumina, and so on, which lose their physisorbed  $H<sub>2</sub>O$  first and then surface hydroxy groups next with increasing temperature. Under the mild dehydration condition at 393 K, most of the water is desorbed, leaving the largest amount of exposed  $H^+$  on H-Hol, on which the largest amount of  $NH<sub>3</sub>$  can be adsorbed irreversibly. By dehydration at 473 K, detachment of water and external H+ sites becomes sufficient to give rise to some coordinately unsaturated (cus) sites on the external surface of <sup>H</sup>-Hol, which can then serve as new dissociative sites for water adsorption. Dehydration at 573 K leads to rearrangement of surface sites on the external surface, namely, the disappearance of cus sites at the external surface while producing some cus sites inside the tunnel structure. The cus sites near the entrance will then be dissociative sites for  $H_2O$  adsorption and  $NH_3$  adsorption sites after rehydration. Finally, dehydration above 673 K leads to sufficient disappearance of lattice protons and partial or complete collapse of the crystal structure, thus bringing about minimal  $NH<sub>3</sub>$  adsorption.

#### **Conclusion**

1. Adsorption results show that  $N_2$ ,  $O_2$ , Ar, CO, and  $CO<sub>2</sub>$  with molecular diameters above 0.33 nm are excluded from the tunnel structure of H-Hol. However,  $H<sub>2</sub>O$  and  $NH<sub>3</sub>$  can be inserted into the tunnel structure with the aid of strong interaction with the surface of <sup>H</sup>-Hol. These results indicate that the effective pore opening of H-Hol for gas adsorption is around 0.265 nm, i.e., the molecular diameter of  $H_2O$ .

2. Dehydration decreases not only the adsorption amount of NH<sub>3</sub> but also that of H<sub>2</sub>O, indicating that  $H^+$ contents in H-Hol play an important role in both NH3 and  $H_2O$  adsorption. Strong interaction between  $NH<sub>3</sub>$ or  $H_2O$  and the H<sup>+</sup> sites contained in H-Hol leads to a slow adsorption process. Time-courses of  $NH<sub>3</sub>$  adsorption show that NH<sub>3</sub> is adsorbed on the external surface or the tunnel entrance of H-Hol at first and then inserted into the inner part of the tunnel; the insertion stage is the more rate-determining process.

3. Adsorption and FT-IR results show that evacuation at a higher temperature leads to dehydration of physisorbed water in concurrence with a stepwise disappearance of proton sites on H-Hol. While one part of the dehydrated sites can become the dissociative sites for  $H_2O$  adsorption and recovered for  $NH_3$  adsorption after rehydration, some proton sites are irreversibly lost due to surface rearrangement by valence change of Mn ions.

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